Polyoxometalates

DOI: 10.1002/anie.200604180

Self-Patterning of Hydrophobic Materials into Highly Ordered Honeycomb Nanostructures at the Air/Water Interface**

Dawei Fan, Xiangfeng Jia, Peiqin Tang, Jingcheng Hao,* and Tianbo Liu*

The patterning of functional materials into well-defined reproducible nanostructures with tunable sizes on incompatible surfaces is a challenging goal of both fundamental and practical importance. Herein, we report the unprecedented self-assembly of hydrophobic double-chain cationic surfactant hybrid materials DODMACl-{Mo₇₂Fe₃₀} at the air/water interface, which leads to the formation of highly ordered honeycomb architectures. DODMACl is dioctadecyldimethy-lammonium chloride, {Mo₇₂Fe₃₀} is the inorganic macroanion Mo^{VI}₇₂Fe^{III}₃₀O₂₅₂L₁₀₂·x H₂O ($x \approx 180$, L=H₂O, CH₃COO⁻, Mo₇Oⁿ⁻_{8/9}).

Recent progress in the synthesis of nanoscale polyoxometalate (POM) molecular clusters, [1] especially polyoxomolybdates and polyoxotungstates with well-defined structures and potential applications in catalysis, electrochemistry, biomedicine, and magnetism, as sensors, surfaces, and coatings, [1c,2] has successfully pushed the size of inorganic molecules to the nanometer scale, and offered new opportunities in many different fields. Some examples of water-soluble POMs are wheel-shaped $\{Mo_{154}\}$, [3] $\{Mo_{176}\}$, [4] and $\{Mo_{248}\}$, [5] C_{60} -like hollow spherical "Keplerate" $\{Mo_{132}\}$ [6] and $\{Mo_{72}Fe_{30}\}$, [7] as well as the hedgehog-type cluster $\{Mo_{368}\}$. [8]

In this study, we use the 2.5-nm-diameter, C_{60} -like molecule $\{Mo_{72}Fe_{30}\}$ (Figure 1 A) as our model system. Many POMs are quite soluble in polar solvents because they contain a large number of water ligands and are negatively charged in solution (similar to polyelectrolytes), with the negative charge balanced by small counterions such as Na⁺ and NH₄⁺. $\{Mo_{72}Fe_{30}\}$ clusters are a little different. They exist as almost neutral molecules in crystals, but in solution they behave like a weak acid: the water ligands attached to the Fe^{III} centers tend to partially deprotonate, thus making $\{Mo_{72}Fe_{30}\}$ clusters slightly negatively charged (carrying several localized charges). The degree of deprotonation depends on the solution pH,^[9] from almost 0 at pH 3.0 to 22 at pH 4.9. The hydrophilic POM surfaces can be adjusted to hydrophobic by forming surfactant-encapsulated

[*] D. Fan, X. Jia, P. Tang, Prof. Dr. J. Hao
Key Laboratory of Colloid and Interface Chemistry of Ministry of
Education, Shandong University
Jinan 250100 (P.R. China)
Fax: (+86) 531-8836-6074
E-mail: jhao@sdu.edu.cn
Prof. Dr. T. Liu
Department of Chemistry, Lehigh University
Bethlehem, PA 18015 (USA)

Bethlehem, PA 18015 (USA) Fax: (+1) 610-758-6536 E-mail: til204@lehigh.edu

[**] This work was supported by the NSFC (20625307, 20571048, 20428101) and NSF (CHM0545983).

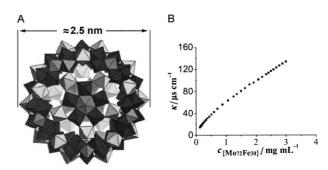


Figure 1. A) Structure of $\{Mo_{72}Fe_{30}\}$ "Keplerate" giant molecule. B) Conductivity data of aqueous solutions of $\{Mo_{72}Fe_{30}\}$ as a function of concentration.

clusters (SECs), through encapsulating the inorganic macroanions with a protective shell of water-soluble cationic species (such as surfactants) via electrostatic forces. [10] Many approaches have been developed to fabricate composite multilayers by electrostatic layer-by-layer self-assembly of POMs modified with a variety of water-soluble cationic species, including transition-metal complexes, cationic surfactants, polycations, and bipolar pyridine. [11] Recently, liquid-crystalline phases formed by ionic self-assembly of POM–surfactant complexes were also prepared. [12]

For a 1.0 mg mL⁻¹ $\{Mo_{72}Fe_{30}\}$ aqueous solution the pH is 3.4, which suggests that each {Mo₇₂Fe₃₀} anion carries about seven net negative charges.^[10d] The conductivity data, shown in Figure 1B as a function of {Mo₇₂Fe₃₀} concentration, clearly confirm that {Mo₇₂Fe₃₀} is a weak electrolyte. Phase transfer of {Mo₇₂Fe₃₀} from aqueous medium (upper phase) to a lower CHCl₃ phase can be achieved with the help of a certain amount of surfactant (DODMACI) dissolved in CHCl₃, which suggests that the concentration of DODMACl in CHCl₃ is critical to determining whether the {Mo₇₂Fe₃₀} macroanions remain in the aqueous or CHCl₃ phase. If enough DODMACl is added, {Mo₇₂Fe₃₀} can be transferred into the organic phase by forming {Mo₇₂Fe₃₀}–DODMA complexes through electrostatic interactions between the {Mo₇₂Fe₃₀} anions and the DODMA⁺ cations. The yellow color of the aqueous phase (from {Mo₇₂Fe₃₀}) continuously fades, while the color of the CHCl₃ phase turns gradually yellow. However, an excess amount of surfactant can form a second layer on the surface of complexes, with their hydrophilic polar heads facing outside, which makes the complexes relatively hydrophilic. Thus, the complexes in the CHCl₃ phase will be partially transferred back into the aqueous phase in the presence of an excess amount of surfactant.

Solvent-cast architectures of {Mo₇₂Fe₃₀}-DODMA complexes in organic medium were prepared at the air/water



interface by evaporating CHCl₃, and were studied by transmission electron microscopy (TEM) using the method reported in the literature.^[10a-c] We originally speculated that the fully hydrophobic complexes should spread homogeneously on the water surface, similar to the case of spreading a layer of oil at the air/water interface. However, in reality we found that the arrangement of complexes was very complicated and dependent on the concentration of DODMACI. When the DODMACl concentration is lower than 1.2 mg mL^{-1} in CHCl₃, the architectures of $\{Mo_{72}Fe_{30}\}$ -DODMA complexes at the air/water interface exhibit organized crystalline structures formed by the packing of the complexes. TEM observations reveal both monolayer (Figure 2A) and multilayer (Figure 2B and C) architectures with three-dimensionally ordered supramolecular structures spread over a broad area. The electron diffraction patterns (Figure 2, right column) confirm that these mono- and multilayer films are composed of complexes.

When the concentration of DODMACI reaches 1.2 mg mL^{-1} in CHCl₃ ($c_{\text{DODMACI}} = 2.02 \text{ mmol L}^{-1}$; molar ratio of $\{\text{Mo}_{72}\text{Fe}_{30}\}$ /surfactant $r \approx 1.90$), unique self-assembly

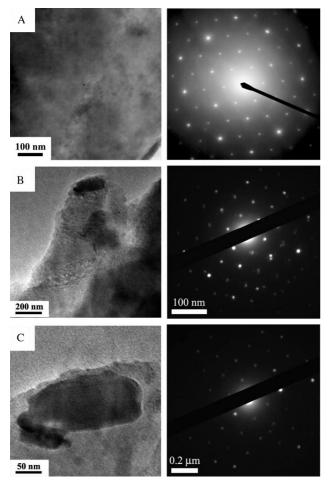


Figure 2. TEM images (left column) and the corresponding electron diffraction patterns (right column) of organized crystalline-structured monolayer films (A) and coexistent mono- and multilayer architectures (B) of {Mo₇₂Fe₃₀}-DODMA complexes. C) High-resolution TEM (HRTEM) image (left) and electron diffraction pattern (right) of organized crystalline-structured multilayer films.

of $\{Mo_{72}Fe_{30}\}$ –DODMA complexes into microsized structures was observed. Typical TEM and HRTEM images are shown in Figure 3, obtained from solvent-cast films of $\{Mo_{72}Fe_{30}\}$ –DODMA complexes at $c_{DODMACI} = 1.2 \text{ mg mL}^{-1}$ in CHCl₃.

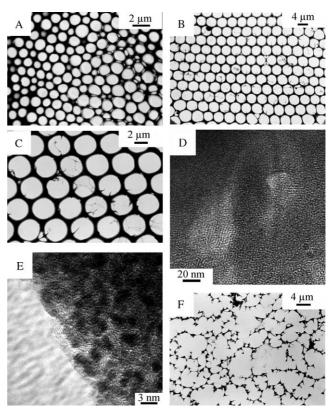


Figure 3. TEM images of A) $\{Mo_{72}Fe_{30}\}$ –DODMA complexes at $c_{DODMACI}=1.2 \text{ mg mL}^{-1} \text{ in CHCl}_3; \text{ B, C}\}$ highly ordered honeycomb architectures of $\{Mo_{72}Fe_{30}\}$ –DODMA complexes. D, E) HRTEM images of the highly ordered honeycombs, which reveal the ordered texture with $\{Mo_{72}Fe_{30}\}$ at $c_{DODMACI}=2.0 \text{ mg mL}^{-1} \text{ in CHCl}_3$. F) TEM image of $\{Mo_{72}Fe_{30}\}$ –DODMA complexes at $c_{DODMACI}=4.0 \text{ mg mL}^{-1} \text{ in CHCl}_3$.

The thickness of the wall between pores is in general smaller than 0.5 µm and the pore sizes are not homogeneous, ranging from 0.5 to 1.5 μ m (Figure 3 A). At $c_{\rm DODMACI} = 2.0 \, \rm mg \, mL^{-}$ $(c_{\text{DODMACI}} = 3.37 \text{ mmol L}^{-1} \text{ and } r \approx 1.150), \text{ the nanostructure}$ formed by {Mo₇₂Fe₃₀}-DODMA complexes (Figure 3B and C) becomes highly ordered and porous, with honeycomb holes of uniform pore size ($\approx 3.5 \, \mu m$). The average thickness of the walls between the pores remains about 0.6 µm. HRTEM studies indicate that the walls are composed of many 2.5-nm-diameter dark objects (Figure 3D and E) that can be attributed to single {Mo₇₂Fe₃₀} clusters. This result confirms that the {Mo₇₂Fe₃₀} clusters play a critical role in constructing the uniform honeycomb structures, together with the surfactant. Scanning electron microscopy (SEM) of the same system confirms the porous structure (Figure 4). In particular, it shows that small holes exist inside the 500-nmthick walls.

When the concentration of DODMACl is higher than $4.0~{\rm mg\,mL^{-1}}$ in CHCl₃ ($c_{\rm DODMACl}\!=\!6.73~{\rm mmol\,L^{-1}}$ and $r\approx1:300$; see Figure 3F), the unique honeycomb structures

Communications

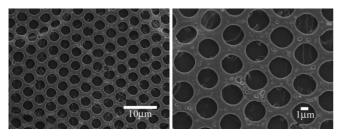


Figure 4. SEM images of highly ordered honeycomb architectures of $\{Mo_{72}Fe_{30}\}$ -DODMA complexes.

are not observed any more. In summary, the formation of highly ordered patterns formed by complexes heavily depends on the surfactant concentration, given the fixed {Mo₇₂Fe₃₀} concentration in aqueous solution. If the amount of surfactant is low (barely enough to cover the POM macroanions and form complexes), regular close-packing of spherical complexes is observed. With increasing surfactant concentration, the free-energy-favored patterning structure gradually becomes the unique honeycomb structure. This structure first appears to have polydispersed pores (Figure 3A), then the pores become more uniform and the packing of pores becomes more ordered with increasing surfactant concentration until an optimum is achieved (Figure 3B-E). In the presence of excess surfactant, the highly ordered patterns become a disordered structure again. This could be because excess surfactant will make the complexes more hydrophilic. The relatively hydrophilic complexes have a higher affinity to the water phase, so they cannot form ordered patterns at the air/water interface.

Figure 5 A shows the ¹H NMR spectrum of DODMACl in CDCl₃ solution. The peaks are very sharp and each group of protons on DODMACl can be readily identified. Figure 5 B shows the ¹H NMR spectrum of $\{Mo_{72}Fe_{30}\}$ –DODMA complex in CDCl₃ at $c_{DODMACl} = 2.0 \text{ mg mL}^{-1}$. The α -CH₂, β -CH₂, and ω '-CH₃ resonances become very weak as a result of the strong electrostatic interaction between the DODMA⁺ cations and $\{Mo_{72}Fe_{30}\}$ anions, which changes the chemical

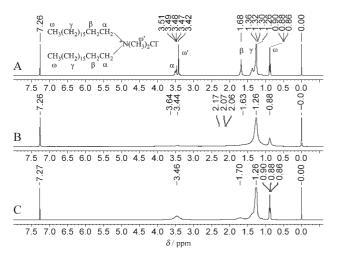


Figure 5. ¹H NMR spectra of A) DODMACI in CDCl₃, and $\{Mo_{72}Fe_{30}\}$ –DODMA in CDCl₃ at B) $e_{DODMACI}=2.0$ and C) 8.0 mg mL⁻¹.

environments of these moieties. In Figure 5C (complexes in CDCl₃ solution at $c_{\text{DODMACI}} = 8.0 \text{ mg mL}^{-1}$), the peaks arising from α -CH₂, β -CH₂, and ω' -CH₃ reappear because of the excess surfactant, but are much broader than those in Figure 5A (on account of the electrostatic interaction between surfactant and $\{Mo_{7}Fe_{30}\}$).

The above results show that the DODMA+ cations encapsulate the $\{Mo_{72}Fe_{30}\}$ anions effectively at $c_{DODMACl}$ = 2.0 mg mL⁻¹. When the hydrophobic {Mo₇₂Fe₃₀}–DODMA complexes are transferred into the air/water interface, they should shrink in volume to minimize the free energy, [13] that is, to minimize surface tension. The shape of the self-assembled structures at the air/water interface should be the optimal topology. In our current case, with $c_{\text{DODMACl}} = 2.0 \text{ mg mL}^{-1}$ in CHCl₃, honeycomb architectures seem to be the optimal morphology with the minimum surface energy. Another possible explanation of the honeycomb structure is that the evaporation of organic solvent (CHCl₃) at the air/water interface leads to the condensation of vapor in air into microscale water beads, which result in micropores after drying, as discovered by François et al.[14] We notice that our current experimental conditions are different, and more studies are needed to elucidate the mechanism of formation of the honeycomb structure. At the moment, we can at least claim that this is a novel way of fabricating highly ordered porous structures at the air/water interface, with incorporation of a large amount of functional inorganic material homogeneously into the walls of the pores.

In summary, the $\{Mo_{72}Fe_{30}\}$ –DODMA complexes produced in CHCl $_3$ form patterns with highly ordered and uniform structures at the air/water interface after evaporation of the organic solvent. The concentration of DODMACl in CHCl $_3$ is critical. We are working on other POMs, such as $\{Mo_{132}\}$, to test the generality of these observations. The formation of highly ordered, uniform, reproducible nanopores at an interface, with homogeneously incorporated inorganic materials, might find various potential applications, such as the development of new materials for catalysis, separation membranes, microstructured electrode surfaces, nanocontainers, or nanoreactors.

Experimental Section

 $\{Mo_{72}Fe_{30}\}$ was synthesized according to the literature. $^{[6,7]}$ A series of solutions of DODMACl (0-20 mg mL⁻¹) in CHCl₃ (5.0 mL) were mixed with an aqueous solution of $\{Mo_{72}Fe_{30}\}\ (1.0 \text{ mg mL}^{-1}, 5.0 \text{ mL}).$ Subsequently, the samples were sonicated at 100 W for 5 min and equilibrated at 25.0 ± 0.1 °C for 1 month. An amount of CHCl₃-phase solution was dropped onto a pure water surface; the solvent-cast architectures of {Mo₇₂Fe₃₀}-DODMA complexes formed after evaporating CHCl₃ were studied by TEM observations.^[10a-c] The highly ordered honeycomb superstructures of {Mo₇₂Fe₃₀}-DODMA complexes in CHCl3 were obtained at an acceleration voltage of 100 kV (JEM-100 CXIII, JEOL). HRTEM images were observed at an acceleration voltage of 200 kV (JEM-2010, JEOL). SEM images were obtained with a JEOL JSM6700F field-emission scanning electron microscope. Three samples were prepared and equilibrated for 1 month at 25.0 ± 0.1 °C: 1) DODMACl in CDCl₃ (2.0 mg mL^{-1}); and $\{Mo_{72}Fe_{30}\}$ (1.0 mg mL⁻¹) in D₂O (5.0 mL) mixed with 2) 2.0 and 3) 8.0 mg mL⁻¹ DODMACl in CDCl₃ (5.0 mL). The organic phase (CDCl₃) was examined by ¹H NMR spectroscopy in 5-mm tubes. A

Bruker AM-400 spectrometer was used for the 1H NMR measurements at 25.0 $\pm\,0.1\,^{\circ}\text{C}.$

Received: October 12, 2006 Revised: December 15, 2006 Published online: March 22, 2007

Keywords: microporous materials · phase transitions · polyoxometalates · self-assembly · surfactants

- a) A. Müller, C. Serain, Acc. Chem. Res. 2000, 33, 2-10; b) A. Müller, M. Henry, C. R. Chim. 2003, 6, 1201-1208; c) L. Cronin in Comprehensive Coordination Chemistry II, Vol. 7 (Eds.: J. A. McCleverty, T. J. Meyer), Elsevier, Amsterdam, 2004, pp. 1-56; d) Polyoxometalate Molecular Science: From Biology to Nanotechnology (Eds.: J. J. Borrás-Almenar, E. Coronado, A. Müller, M. T. Pope), Kluwer, Dordrecht, 2003.
- [2] N. Casañ-Pastor, P. Gómez-Romero, Front. Biosci. 2004, 9, 1759-1770.
- [3] A. Müller, S. K. Das, V. P. Fedin, E. Krickemeyer, C. Beugholt, H. Bögge, M. Schmidtmann, B. Hauptfleisch, Z. Anorg. Allg. Chem. 1999, 625, 1187–1192.
- [4] A. Müller, M. Koop, H. Bögge, M. Schmidtmann, C. Beugholt, Chem. Commun. 1998, 15, 1501–1502.
- [5] A. Müller, S. Q. N. Shah, H. Bögge, M. Schmidtmann, *Nature* 1999, 397, 48–50.
- [6] A. Müller, E. Krickemeyer, H. Bögge, M. Schmidtmann, F. Peters, Angew. Chem. 1998, 110, 3567–3571; Angew. Chem. Int. Ed. 1998, 37, 3359–3363.
- [7] A. Müller, S. Sarkar, S. Q. N. Shah, H. Bögge, M. Schmidtmann, S. Sarkar, P. Kögerler, B. Hauptfleisch, A. X. Trautwein, V.

- Schünemann, Angew. Chem. 1999, 111, 3435–3439; Angew. Chem. Int. Ed. 1999, 38, 3238–3241.
- [8] a) A. Müller, E. Beckmann, H. Bögge, M. Schmidtmann, A. Dress, Angew. Chem. 2002, 114, 1210–1215; Angew. Chem. Int. Ed. 2002, 41, 1162–1167; b) A. Müller, B. Botar, S. K. Das, H. Bögge, M. Schmidtmann, A. Merca, Polyhedron 2004, 23, 2381–2385.
- [9] T. Liu, B. Imber, E. Diemann, G. Liu, K. Cokleski, H. Li, Z. Chen, A. Müller, J. Am. Chem. Soc. 2006, 128, 15914–15920.
- [10] a) D. G. Kurth, P. Lehmann, D. Volkmer, H. Cölfen, M. J. Koop, A. Müller, A. Du Chesne, Chem. Eur. J. 2000, 6, 385-393; b) D. Volkmer, A. Du Chesne, D. G. Kurth, H. Schnablegger, P. Lehmann, J. Koop, A. Müller, J. Am. Chem. Soc. 2000, 122, 1995-1998; c) D. G. Kurth, P. Lehmann, D. Volkmer, A. Müller, D. Schwahn, J. Chem. Soc. Dalton Trans. 2000, 3989-3998; d) T. Liu, J. Cluster Sci. 2003, 14, 215-226; e) W. Bu, H. Li, H. Sun, S. Yin, L. Wu, J. Am. Chem. Soc. 2005, 127, 8016-8017; f) W. Bu, H. Li, W. Li, L. Wu, C. Zhai, Y. Wu, J. Phys. Chem. B 2004, 108, 12776-12782.
- [11] a) S. Liu, D. Volkmer, D. G. Kurth, J. Cluster Sci. 2003, 14, 405–419; b) D. G. Kurth, D. Volkmer, M. Ruttorf, B. Richter, A. Müller, Chem. Mater. 2000, 12, 2829–2831.
- [12] a) T. Zhang, C. Spitz, M. Antonietti, C. F. J. Faul, *Chem. Eur. J.* 2005, 11, 1001–1009; b) S. Polarz, B. Smarsly, M. Antonietti, *ChemPhysChem* 2001, 2, 457–461.
- [13] M. Srinivasarao, D. Collings, A. Philips, S. Patel, *Science* 2001, 292, 79–83.
- [14] a) G. Widawski, B. François, *Nature* 1994, 369, 387–389; b) B. François, O. Pitois, J. François, *Adv. Mater.* 1995, 7, 1041–1044; c) O. Pitois, B. François, *Eur. Phys. J. B* 1999, 8, 225–231; d) O. Pitois, B. François, *Colloid Polym. Sci.* 1999, 277, 574–578.